# **Vanadosilicate Mesoporous SBA-15 Molecular Sieves Incorporated with** *N-***Alkylphenothiazines**

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Mesoporous silica SBA-15 molecular sieves have been incorporated with various amounts of vanadium via incipient-wetness impregnation with vanadium oxide triisopropoxide in hexane followed by calcination. The products (VSBA-15) have been characterized by nitrogen adsorption and diffuse reflectance UV-visible and electron spin resonance spectroscopies. The results show that the vanadium exists as isolated tetrahedral  $V<sup>5+</sup>$  ions in dehydrated VSBA-15 with Si/V  $\geq$  40 while vanadium chains develop as vanadium loading increases. Upon hydration a large fraction of the vanadium is transformed to square pyramidal and distorted octahedral  $\check{V}^{5+}$  ions by additional coordination to water molecules, suggesting an interior wall surface location for these  $\mathrm{V}^{5+}$  ions. Upon CO reduction or incorporation of *N*-alkylphenothiazines (PC<sub>n</sub>), part of the  $V^{5+}$  is reduced to vanadyl  $VO^{2+}$  ions. While incorporation of  $PC_n$  into VSBA-15 in the dark generates some  $PC_n^+$  cation radicals, ultraviolet photoionization at room temperature further increases the yield for  $Si/V = 80$ and 40. The PC*<sup>n</sup>* <sup>+</sup> radicals are more stable in VSBA-15 than in siliceous SBA-15, and the photoyield increases with vanadium loading. The photoyield and stability of  $PC_n^+$  also increase with increasing alkyl chain length of the *N-*alkylphenothiazines.

### **Introduction**

Transition-metal-substituted molecular sieves are of current interest as catalysts for oxidation of a variety of organic compounds where effective but toxic stoichiometric oxidants are used. $1,2$  These materials possess uniform pore structure and high surface area and are effective with "clean" and economical oxygen donors such as  $30\%$  aqueous hydrogen peroxide.<sup>1-3</sup> Titanium<sup>3,4</sup> and vanadium-substituted silicalites<sup>1,5</sup> are among the most successful materials. Recently, various transitionmetal-substituted mesoporous MCM-41 and MCM-48 molecular sieves have also been produced. $6-9$  These materials possesses uniform mesopore channels varying from about 15 Å to about 100 Å and have a clear advantage over smaller channel microporous molecular sieves for the adsorption and transformation of large organic molecules.6-<sup>10</sup>

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Photoinduced charge separation in various artificial photoredox systems has long been studied for solar energy conversion and storage.<sup>11,12</sup> The charge separation in these systems generally involves a redox reaction between a photosensitive electron donor and a suitable electron acceptor, which generates a pair of radical ions.13,14 Tremendous effort has been made to extend the lifetime of the photogenerated radical ions in order to exploit the associated energy prior to back electron transfer.13-<sup>19</sup> Generally, long-lived charge separation can be achieved in heterogeneous media such as silica  $gels<sup>17,18</sup>$  and zeolites<sup>19</sup> where the radical ion pairs can be sterically separated and molecular mobility is restricted.

SBA-15 is a newly discovered mesoporous silica molecular sieve with uniform tubular channels variable from 50 to 300 Å.20,21 Incorporation of transition metal ions into SBA-15 by direct hydrothermal synthesis appears unlikely since this mesoporous phase is syn-(1) Bellussi, G.; Rigutto, M. S. In *Advanced Zeolite Science and* **thesized in strong acidic media (2 M HCl).**<sup>20,21</sup> But

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transition metal ion incorporation might be achieved by "postsynthesis" grafting involving reaction with a transition metal ion solution.<sup>21-24</sup> We have recently reported such an approach to incorporate titanium<sup>25</sup> or aluminum21 into SBA-15 as framework species and to prepare thin  $TiO<sub>2</sub>$  molecular films anchored inside the mesopores of SBA-15.25 Here we report a preparation of SBA-15 with incorporated vanadium by reacting SBA-15 with dry hexane solutions of vanadium alkoxide.<sup>24</sup> The products are characterized by nitrogen adsorption and diffuse reflectance UV-visible and electron spin resonance (ESR) spectroscopy. These materials are used as heterogeneous hosts for methylphenothiazine and *N*alkylphenothiazines which are photosensitive electron donors. Photoionization of these *N-*alkylphenothiazine molecules is achieved at room temperature with 320 nm light.

#### **Experimental Section**

**Sample.** A synthesis procedure for silica SBA-15, designated in this work as SiSBA-15, has been reported.<sup>20,21</sup> Prior to vanadium loading the SiSBA-15 was dehydrated in air at 550 °C for 12 h and then transferred to a glovebox under flowing nitrogen. Vanadium-incorporated SBA-15, designated as VSBA-15, was prepared in a shallow dish inside the glovebox. Vanadium was loaded into SiSBA-15 at room temperature by incipient-wetness impregnation.<sup>24</sup> For every 1 g of SiSBA-15, varying amounts of vanadium oxide triisopropoxide (Aldrich) in 10 g of hexane were used. The vanadium concentration in the solution varies from 0.05 to 0.50 M depending on the desired vanadium loading. The impregnated material was stored in nitrogen at room temperature for 12 h and dried by evaporation in a vacuum. Calcination was performed in air at 550 °C for 5 h. These samples are designated as VSBA-15-(*x*), where *x* is the stoichiometric Si/V molar ratio from the incipient-wetness impregnation. For comparison,  $V_2O_5$  powder was used as received from Aldrich.

For CO reduction, VSBA-15 samples were loaded into 3 mm o.d. by 2 mm i.d. Suprasil quartz tubes, evacuated at 425 °C for 2 h, and then exposed to 50 Torr of CO at 200 °C for 2 h. The CO-reduced VSBA-15 samples were flame-sealed for ESR measurement.

Methylphenothiazine was used as received from Aldrich. *N*-Alkylphenothiazines (PC<sub>*n*</sub>,  $n = 1$ , 6, 10, 16, where *n* is the number of the carbon atoms in the alkyl chain) were synthesized following a reported procedure.<sup>14,26</sup> PC<sub>n</sub>-loaded VSBA-15, designated as PC*n*/VSBA-15, was prepared inside a glovebox with flowing nitrogen in the dark. A typical procedure is as follows. Dehydrated VSBA-15 (0.25 g) was immersed in 10 mL of 0.01 M  $\overline{PC}_n$  in benzene for 0.5 h. The benzene solvent was then removed by flowing nitrogen gas directly on the surface of the solution for 2 h. The dried PC*n*/VSBA-15 powder was loaded in flowing nitrogen into 3 mm o.d. by 2 mm i.d. Suprasil quartz tubes and sealed with Parafilm for photoionization and ESR measurement.

**Photoionization**. A Cermax 150 W xenon lamp (ILC-LX150F) was used. The light was passed through a 10 cm water filter and a Corning 7-54 glass filter to give 320 nm irradiation of  $1.3 \times 10^6$  ergs cm<sup>-2</sup>/s in the range from 240 to 400 nm.14,18 Each PC*n*/VSBA-15 sample was irradiated in a quartz dewar that was rotated at 4 rpm to ensure even irradiation.

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**Table 1. Pore Structure Parameters of VSBA-15 Calculated from the Desorption Branch of Nitrogen Adsorption Isotherms Using the Barrett**-**Joyner**-**Halenda Formula**

materials	$A_{\mathrm{BFT}}$ $\binom{m^2}{m^2}$ g <sup>-1</sup> )	$A_{\rm{B\,IH}}$ $(m^2 g^{-1})$	$V_{\rm{B\,IH}}$ $\rm (cm^3\,g^{-1})$	$D_{\rm BJH}$ (Å)
$SiSBA-15$	997	926	1.17	51
<b>VSBA-15-(80)</b>	835	789	1.02	52
VSBA-15-(40)	817	802	1.05	52
VSBA-15-(20)	582	621	0.82	53
VSBA-15-(10)	551	564	0.74	52

**Instrumentation.** Nitrogen adsorption isotherms were measured at 77 K using a Micromeritics Gemini 2375 analyzer. The volume of adsorbed  $N_2$  was normalized to standard temperature and pressure. Prior to the experiments, samples were dehydrated at 250 °C for 5 h. The specific surface area, *A*BET, was determined from the linear part of the BET equation  $(P/P<sub>0</sub> = 0.05-0.31)$ . Although the Barret-Joyner-Halenda (BJH) analysis underestimates the pore size, $28-31$  relative changes in the pore size are accurately portrayed. The calculation of the pore size distribution was performed using the desorption branches of the  $N_2$  adsorption isotherms and the BJH formula.<sup>27</sup> The cumulative pore surface area, A<sub>BJH</sub>, and volume,  $V_{\text{BJH}}$ , were obtained from the pore size distribution curves. The average pore diameter,  $D_{\text{BJH}}$ , was calculated as  $4V_{\text{BJH}}/A_{\text{BJH}}$ .

UV-vis spectra were measured with a Perkin-Elmer 330 spectrophotometer equipped with a 60 mm Hitachi integrating sphere accessory. Powder samples were loaded in a quartz cell with Suprasil windows, and spectra were collected in the 200- 1000 nm wavelength range against a siliceous SBA-15 standard.

ESR spectra were recorded at X-band at ambient temperature on a Bruker ESP 300 spectrometer. The magnetic field was calibrated with a Varian E-500 gaussmeter. The microwave frequency was measured by a Hewlett-Packard HP 5342A frequency counter.

#### **Results**

**Nitrogen Adsorption.** Characterization of siliceous SiSBA-15 by low-angle X-ray diffraction and nitrogen adsorption has been reported elsewhere,<sup>21,25</sup> which confirms the uniformity of the mesoporous structure (Table 1). VSBA-15 gives similar  $N_2$  adsorption isotherms to that of SiSBA-15 but with decreasing overall  $N_2$  adsorption volume as the vanadium loading increases. Figure 1 shows a typical adsorption isotherm for VSBA-15-(40), which shows an irreversible type IV adsorption isotherm32 with a sharp inflection in the *P*/*P*<sup>0</sup> range from 0.60 to 0.80 characteristic of capillary condensation within uniform pores. A corresponding BJH plot leads to a rather narrow pore size distribution with a pore size of 52 Å (Figure 1, insert). The calculated BET specific surface area is around 1000  $\mathrm{m}^2/\mathrm{g}$  for SiSBA-15, and it slightly decreases with increasing vanadium loading up to  $Si/V = 40$  (Table 1). Further

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**Figure 1.** Adsorption isotherms of nitrogen in VSBA-15-(40) at 77 K. The insert shows the corresponding pore size distribution curve.

increase in vanadium loading greatly reduces the BET specific surface area (Table 1). The mesopore surface area *A*BJH around 900 m2/g for SiSBA-15 also decreases with vanadium loading. However, the average mesopore diameter, which is about 51 Å for SiSBA-15, remains about the same as the vanadium loading increases up to  $Si/V = 10$  (Table 1). The pore size distributions all look similar for any vanadium loading.

**UV**-**Vis Spectroscopy**. Dehydrated VSBA-15 materials are white. At low vanadium loading corresponding to  $Si/V = 80$ , VSBA-15 shows only one broad band with a maximum near 270 nm (Figure 2a), which can be assigned to low-energy charge-transfer transitions between tetrahedral oxygen ligands and a central  $V^{5+}$ ion.7,9,33,34 As the vanadium loading increases up to Si/V  $=$  10, the maximum absorption shifts near 300 nm (Figure 2b-d). None of the spectra match that of dehydrated  $V_2O_5$  (Figure 2e) so  $V_2O_5$  does not seem to be formed.

Upon hydration in air the color of VSBA-15 changes rapidly to bright yellow and then orange as reflected by broadening of the UV-vis spectrum toward lower energy (Figure 3). Hydrated VSBA-15 at low vanadium loading gives a band near 370 nm (Figure 3a,b). Further increase in vanadium loading increases the absorption at 370 nm and broadens the spectrum toward lower energy (Figure 3c,d). Repeated dehydration and exposure to hydrated air shows that both the color and UVvis spectral changes of VSBA-15 are reversible. The VSBA-15 spectra are always different from that of  $V_2O_5$ so  $V_2O_5$  does not seem to be formed.

ESR. Freshly prepared VSBA-15 materials give no ESR signals. Following thermal treatment and subse-



Figure 2. UV-vis spectra (vs siliceous SBA-15) of dehydrated VSBA-15: (a) VSBA-15-(80); (b) VSBA-15-(40); (c) VSBA-15- (20); (d) VSBA-15-(10); (e)  $V_2O_5$ .



Figure 3. UV-vis spectra (vs siliceous SBA-15) of hydrated VSBA-15: (a) VSBA-15-(80); (b) VSBA-15-(40); (c) VSBA-15- (20); (d) VSBA-15-(10); (e)  $V_2O_5$ .

quent CO reduction, all VSBA-15 materials change color from orange to white and then to green and give ESR signals at room temperature as shown in Figure 4. For VSBA-15 with low vanadium loading the ESR spectra

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**Figure 4.** ESR spectra at room temperature of VSBA-15 following CO reduction at 200 °C for  $\overline{2}$  h: (a) SiSBA-15; (b) VSBA-15-(80); (c) VSBA-15-(40); (d) VSBA-15-(20); (e) VSBA-15-(10).

(Figure 4b,c) can be characterized by an axially symmetric set of eight lines originating from vanadyl  $VO^{2+}$ species coupled to its own nuclear spin  $(^{51}V, I_n = 7/2$ , natural abundance 99.8%).35-<sup>37</sup> The spin Hamiltonian parameters are *g*| = 1.937, *A*||= 192 G, *g*<sub>⊥</sub> = 1.984, and  $A_{\perp}$  = 69 G, where *A* is the hyperfine coupling. At higher vanadium loading, the ESR spectra show reduced resolution of the hyperfine structure (Figure 4d,e).

As shown in Figures 5b,c and 6, incorporation of either methylphenothiazine or *N-*alkylphenothiazines with longer alkyl chains into VSBA-15-(80) and VSBA-15-(40) also generates ESR signals of vanadyl  $VO^{2+}$ species in addition to a narrow line at  $g = 2.003$  from phenothiazine radical ions (see below). However, the ESR spectra of  $PC_1/VSBA-15-(20)$  and  $PC_1/VSBA-15-$ (10) with high vanadium loading show mainly  $VO^{2+}$ lines (Figure 5 d,e).

**Photoionization.** Prior to UV irradiation, siliceous PC1/SiSBA-15 does not give any ESR signal (Figure 5a). In contrast, the  $PC_1/VSBA-15-(80)$  and  $-(40)$  materials with vanadium indicate a dark reaction as shown by a quite intense line at  $g = 2.003$  (Figure 5b,c) which is assigned to methylphenothiazine cation radicals  $PC_1^{+,14,18,38-42}$  Since the line from  $PC_1^{+}$  is overlapped

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**Figure 5.** ESR spectra at room temperature of PC<sub>1</sub>/VSBA-15 before UV irradiation: (a)  $PC<sub>1</sub>/SisBA-15$ ; (b)  $PC<sub>1</sub>/VSBA-15$ 15-(80); (c) PC1/VSBA-15-(40); (d) PC1/VSBA-15-(20); (e) PC1/ VSBA-15-(10).

with that from  $VO^{2+}$ , a direct measurement of the signal intensity for  $PC_1^+$  by double integration is not reliable. However, the relative intensity of this line can be estimated from its height, *H*, defined in Figure 5b. This seems reliable since the  $g = 2.003$  is so sharp and intense compared to the multiline  $VO^{2+}$  signal. At higher vanadium content with  $Si/V = 20$  this is not so clear so we do not quantitate these systems.

Following UV irradiation at room temperature the ESR signal intensity *H* of the line at  $g = 2.003$  is also measured at room temperature. The results are given in Figure 7 for both  $PC_1/SiSBA-15$  and  $PC_1/VSBA-15$ materials. The ESR signal of  ${PC_1}^+$  increases greatly during the first 10 min and then approaches a plateau. The photoyield is also dependent on the vanadium content in the materials. The ESR signal intensity of  $PC_1^+$  increases with increasing vanadium content to  $Si/V = 40$ . After UV irradiation at room temperature for 60 min, the samples were stored in the dark to allow  $PC_1^+$  to decay. Figure 8 shows the normalized ESR signal intensities of  ${PC_1}^+$  measured at room temperature following the decay as a percentage of the intensity before decay. Figure 8 shows that the  ${PC_1}^+$  ESR signal intensity for siliceous  $PC<sub>1</sub>/SiSBA-15$  decays relatively rapidly and is reduced to 50% in 1 h and to almost 10% in 10 h. In contrast, the  ${PC_1}^+$  ESR signal intensity from  $PC<sub>1</sub>/VSBA-15-(80)$  and  $PC<sub>1</sub>/VSBA-15-(40)$  containing vanadium decays significantly slower and may show a decreasing decay rate with increasing vanadium content

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 $200<sub>G</sub>$ 

**Figure 6.** ESR spectra at room temperature of PC*n*/VSBA-15-(40) before UV irradiation: (a)  $\overline{PC_1/SiSBA-15}$ ; (b)  $\overline{PC_6/}$ VSBA-15-(80); (c) PC10/VSBA-15-(40); (d) PC16/VSBA-15-(20).



**Figure 7.** Dependence of the ESR signal intensity of  $PC_1{}^+$  in PC1/VSBA-15 versus irradiation time at room temperature: (O)  $PC_1/SiSBA-15$ ; ( $\triangle$ )  $PC_1/VSBA-15-(80)$ ; ( $\nabla$ )  $PC_1/VSBA-15-$ (40); ( $\Diamond$ ) PC<sub>1</sub>/VSBA-15-(20); () PC<sub>1</sub>/VSBA-15-(10).

(Figure 8). After decay for 90 h about 50% of the  $PC_1^+$ is still retained for the vanadium materials.

For PC*n*/VSBA-15-(40) with *N-*alkylphenothiazines having longer alkyl chains, ESR shows a quite intense line prior to UV irradiation (Figure 6). This line at  $g =$ 2.003 is assigned to *N-*alkylphenothiazine cation radicals  ${\rm PC}_n^{++,14}$  The ESR signal intensity  $H$  of  ${\rm PC}_n^+$  radicals increases with increasing alkyl chain length from methyl  $(n = 1)$  to decyl  $(n = 10)$  and then declines with further increase to hexadecyl  $(n = 16)$  (Figure 6). UV irradiation further increases the yield of PC $_1^+$  and PC $_6^+$ radicals by about 45% but has little effect on the yield for  $PC_{10}$ <sup>+</sup> and  $PC_{16}$ <sup>+</sup>. As shown in Figure 9, the ESR signal intensity of  ${PC}_n^+$  decays less as the alkyl chain length increases.



**Figure 8.** Decay of the ESR signal intensity of  $PC_1^+$  in  $PC_1/$ VSBA-15 at room temperature: (O)  $PC_1/SiSBA-15$ ; ( $\triangle$ )  $PC_1/$ VSBA-15-(80); ( $\triangledown$ ) PC<sub>1</sub>/VSBA-15-(40); ( $\diamond$ ) PC<sub>1</sub>/VSBA-15-(20);  $(1) PC<sub>1</sub>/VSBA-15-(10)$ . The insert shows the initial decay period.



**Figure 9.** Decay of the ESR signal intensity of PC*<sup>n</sup>* <sup>+</sup> radicals in PC<sub>n</sub>/VSBA-15-(40) at room temperature: (O) PC<sub>1</sub>/VSBA-15-(40); ( $\triangle$ ) PC<sub>6</sub>/VSBA-15-(40); ( $\nabla$ ) PC<sub>10</sub>/VSBA-15-(40); ( $\diamond$ ) PC<sub>16</sub>/ VSBA-15-(40). The insert shows the initial decay period.

#### **Discussion**

**Pore Structure.** As reported earlier, <sup>21,25</sup> siliceous SBA-15 material possesses a narrow pore size distribution with an average pore size of about 51 Å and a high mesopore surface area  $A_{\text{BJH}}$  of 997 m<sup>2</sup>/g (Table 1). Incorporation of vanadium does not much affect the original pore structure of the parent SBA-15 for vanadium loading lower than  $Si/V = 40$  (Figure 1, Table 1). Further increase in vanadium loading reduces the mesopore surface area (Table 1). Since the calculated mesopore diameter remains almost constant (Table 1), the decrease in surface area might be due to pore blocking.

**Incorporation of Vanadium.** Dehydrated VSBA-15 materials with lower vanadium loading than  $Si/V =$ 40 are white and give only one broad UV-vis band with a maximum at 270 nm (Figure 2). Such a band has been extensively reported for various crystalline materials, such as  $\check{VS}$ -1, VS-2,<sup>1,5,36</sup> VAPO-5,<sup>1,43</sup> and amorphous mesoporous materials<sup>7,9</sup> and has been assigned to colorless tetrahedrally coordinated  $V^{5+}$  ions as monomers and is so assigned in this work. Upon CO reduction or incorporation of methylphenothiazine molecules, these VSBA-15 materials with lower vanadium loading give an ESR signal with hyperfine characteristic of vanadyl  $VO<sup>2+</sup>$  ions in square pyramidal coordination (Figures 4) and 5).<sup>1,5,36,43</sup> This along with the color change of VSBA-15 from white to green indicates the reduction of some  $V^{5+}$  to  $VO^{2+}$  ions. The hyperfine resolution suggests that the  $VO^{2+}$  ions are isolated. The UV-vis results support that the vanadium is incorporated into SBA-15 as isolated species for vanadium loading  $Si/V \ge 40$ .

Upon hydration, VSBA-15 materials with lower vanadium loading give spectra different from those of corresponding dehydrated materials. A band with a maximum at 370 nm is observed in addition to the 270 nm band seen in dehydrated material (Figure 3). In addition, the color of the materials changes from white to bright yellow and then orange during hydration. It is apparent that a fraction of the vanadium ions is accessible to water molecules and can achieve higher coordination numbers than four upon hydration. This implies that these vanadium ions are coordinated to the interior wall surfaces of SBA-15.

With increasing vanadium loading from  $Si/V = 40$  to 10, the UV-vis spectra of dehydrated VSBA-15 materials show a shift of the maximum of the 270 nm band to near 300 nm but do not match that from dehydrated  $V<sub>2</sub>O<sub>5</sub>$  (Figure 2). This is similar to reported spectra from various vanadium-containing materials, where a similar band with a maximum near 300 nm has been observed and ascribed to colorless tetrahedrally coordinated  $\mathrm{V}^{5+}$ ions as chains of tetrahedral vanadia.36 Upon hydration, UV-vis spectra for  $Si/V = 20$  to 10 show that the band for  $V^{5+}$  ions also broadens toward lower energy consistent with increasing coordination with water molecules to become five- and six-coordinated.7,16,36,37 This previous work also does not suggest the formation of  $V_2O_5$ .

**Photoionization.** PC<sub>1</sub>/VSBA-15 materials give a quite intense ESR signal at  $g = 2.003$  at room temperature before irradiation (Figure 5). The ESR signal is typical of  $PC_1^+$  as previously reported for  $PC_1$  in homogeneous solution and in silica gels.38-<sup>42</sup> This indicates that some  ${PC_1}^+$  radicals are produced during the preparation process of the  $PC_1 / VSBA-15$  materials. Ultraviolet irradiation at room temperature generates more  $PC_1$ <sup>+</sup> in siliceous  $PC_1/SiSBA-15$  than in  $PC_1/VSBA-$ 15 materials (Figure 7) which confirms the photooxida-

tion of  $PC<sub>1</sub>$  in both siliceous SiSBA-15 and vanado VSBA-15 materials by UV irradiation at room temperature.

For siliceous  $PC<sub>1</sub>/VSBA-15$  hydroxyl groups on the wall surfaces<sup>25</sup> might be adsorption sites for  $PC_1$  as well as electron acceptors as has previously been suggested for silica gel materials.14,18 These hydroxyl groups are weak electron acceptors and cannot induce the oxidation of PC<sub>1</sub> to PC<sub>1</sub><sup>+</sup> without UV irradiation (Figure 5). For  $PC_1/VSBA-15$  materials a second ESR signal of  $VO^{2+}$  is observed prior to UV irradiation in addition to the ESR signal of  $PC_1^+$  (Figure 7). Thus,  $PC_1^+$  is apparently generated by interaction with  $V^{5+}$  in VSBA-15. Oxidation of  $PC_1$  by  $V^{5+}$  seems to be confirmed by the observation that vanadium(V) oxide triisopropoxide in hexane changes color from yellow to light pink upon addition of PC<sub>1</sub> to the solution. The VO<sup>2</sup> and PC<sub>1</sub><sup>+</sup> yields increase with vanadium content up to  $Si/V = 40$ .

 ${ \mathrm{PC}_1}^+$  generated in SiSBA-15 is relatively unstable at room temperature and decays to about 50% in 1 h and 10% in 10 h (Figure 8), which is comparable to that reported for PC $_{\rm 1}^+$  in silica gel. $^{17,18}$  In comparison, PC $_{\rm 1}^+$ is more stable in VSBA-15 at room temperature and decays at room temperature by only 50% in several days (Figure 8). This seems related to a higher stability for  $VO^{2+}$  and slower back electron transfer to  $PC_1^+$ . The slower decay of  ${PC_n}^+$  with longer alkyl chain lengths is not well understood.

## **Conclusions**

Vanadium has been incorporated into mesoporous silica SBA-15 molecular sieve. Characterization shows that the vanadium is predominantly present as isolated vanadium ions coordinated to the silica walls at vanadium loadings up to  $Si/V = 40$ . Incorporation of Nalkylphenothiazines into these vanadium containing materials generates *N-*alkylphenothiazine cation radicals, which increase in yield further upon ultraviolet irradiation. The photoyield and stability of the *N*alkylphenothiazine cation radicals increase with increasing vanadium loading in SBA-15 and with increasing alkyl chain length.

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